

# Non-degenerate n-type doping by hydrazine treatment in metal work function engineered WSe<sub>2</sub> field-effect transistor

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## Abstract

We report a facile and highly effective n-doping method using hydrazine solution to realize enhanced electron conduction in a WSe<sub>2</sub> field-effect transistor (FET) with three different metal contacts of varying work functions—namely, Ti, Co, and Pt. Before hydrazine treatment, the Ti- and Co-contacted WSe<sub>2</sub> FETs show weak ambipolar behaviour with electron dominant transport, whereas in the Pt-contacted WSe<sub>2</sub> FETs, the p-type unipolar behaviour was observed with the transport dominated by holes. In the hydrazine treatment, a p-type WSe<sub>2</sub> FET (Pt contacted) was converted to n-type with enhanced electron conduction, whereas highly n-doped properties were achieved for both Ti- and Co-contacted WSe<sub>2</sub> FETs with on-current increasing by three orders of magnitude for Ti. All n-doped WSe<sub>2</sub> FETs exhibited enhanced hysteresis in their transfer characteristics, which opens up the possibility of developing memories using transition metal dichalcogenides.

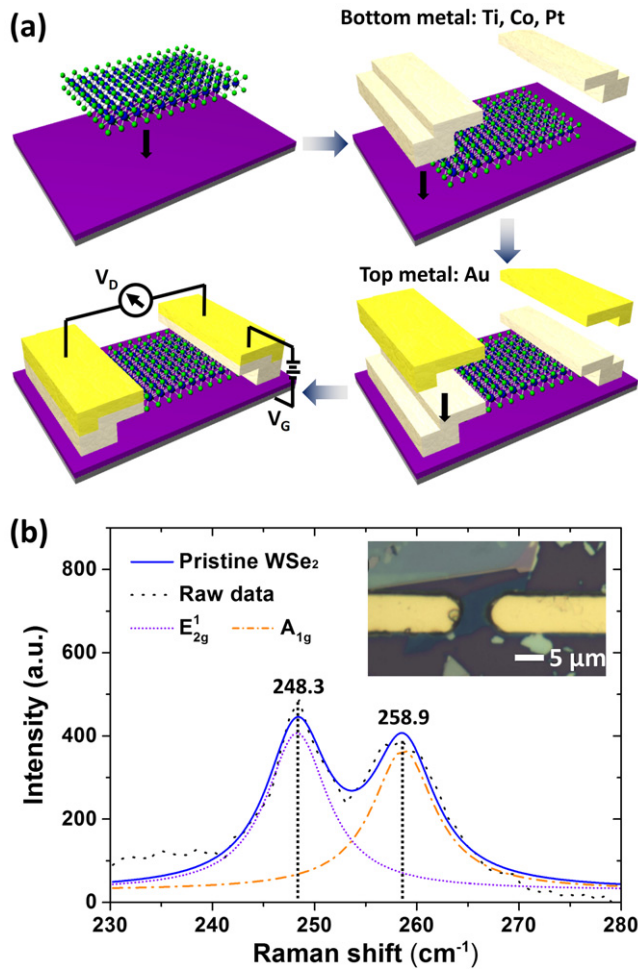
Keywords: WSe<sub>2</sub>, n-doping, hydrazine treatment, metal work function, hysteresis

## 1. Introduction

For the past decade, graphene, a two-dimensional (2D) material, has been widely studied to realize fast and atomically thin field-effect transistors (FET) with unique electrical and mechanical properties such as high mobility, mechanical strength, and transparency and flexibility [1–3]. Inspired by graphene, new 2D materials based on transition metal dichalcogenides (TMDC) have been investigated owing to their unique layer-dependent physical properties. By varying the thickness of TMDCs, it is possible to tune their band gap and optical response as well as improve their on/off ratio [4, 5]. Another tremendous advantage of TMDCs over graphene is the availability of both n- and p-type materials in native forms—e.g., n-type MoS<sub>2</sub> and p-type WSe<sub>2</sub>—which are highly desirable for complementary electronics [6]. However, one of the crucial challenges in realizing high-performance TMDC-based FETs is to obtain good ohmic contact and control the

doping state in these materials. In the past, several metals have been studied for contact engineering in TMDC-based FETs [7]. Further, various doping techniques for TMDCs were used such as metal ion decorated DNA treatment (p-doping) [8], silicon nitride passivation (n-doping) [9], and exposure to nitrogen dioxide (p-doping) [10] and potassium (n-doping) vapour [11]. However, a combined study of the doping effect and contact engineering, which would provide a better understanding of the device operation in TMDC-based electronics with various metals, is still missing.

In this work, we fabricated few-layer WSe<sub>2</sub> FETs with different contact metals (Ti, Co, and Pt) with significant differences in work function and investigated the chemical doping effect by hydrazine solution. Our n-doping process by the dipping method in solution is facile and simple compared to other methods [8–11]. The results show that for Ti- and Co-contacted FETs, hydrazine treatment makes them strongly n-type, and for Pt-contacted FETs, the pristine p-type was converted to n-type.



**Figure 1.** (a) Schematic of the WSe<sub>2</sub> FET fabrication process. (b) Raman spectrum of a thin WSe<sub>2</sub> layer used in this study. Inset shows the optical image of a fabricated WSe<sub>2</sub> FET.

## 2. Experimental details

Figure 1(a) shows schematics of the WSe<sub>2</sub> FET fabrication process. The WSe<sub>2</sub> thin layer was exfoliated from bulk and deposited on 90 nm thick SiO<sub>2</sub>/Si substrate. The electrodes were patterned by optical lithography. A 20 nm thick layer of Ti, Co, or Pt was first deposited as a contacting metal layer on the patterned WSe<sub>2</sub> layer, followed by 50 nm thick Au for an electrode pad. The inset in figure 1(b) shows the optical image of a fabricated WSe<sub>2</sub> FET. The channel length and width of the fabricated devices were approximated by electrode gap size and metal electrode width—i.e., 5 μm and 5 μm, respectively. All electrical characterisations were performed by three-terminal measurement with the channel current controlled using the back gate. In the following discussions, each device with a specific contacting metal is designated as Ti–WSe<sub>2</sub>, Co–WSe<sub>2</sub>, or Pt–WSe<sub>2</sub>. The WSe<sub>2</sub> layer on SiO<sub>2</sub>/Si substrate was identified by Raman spectroscopy, as shown in figure 1(b). The peak positions of the in-plane mode (E<sub>2g</sub><sup>1</sup> at 248.3 cm<sup>-1</sup>) and out-of-plane mode (A<sub>1g</sub> at 258.9 cm<sup>-1</sup>) are in good agreement with the previous report on thin WSe<sub>2</sub> layers [12].

## 3. Results and discussion

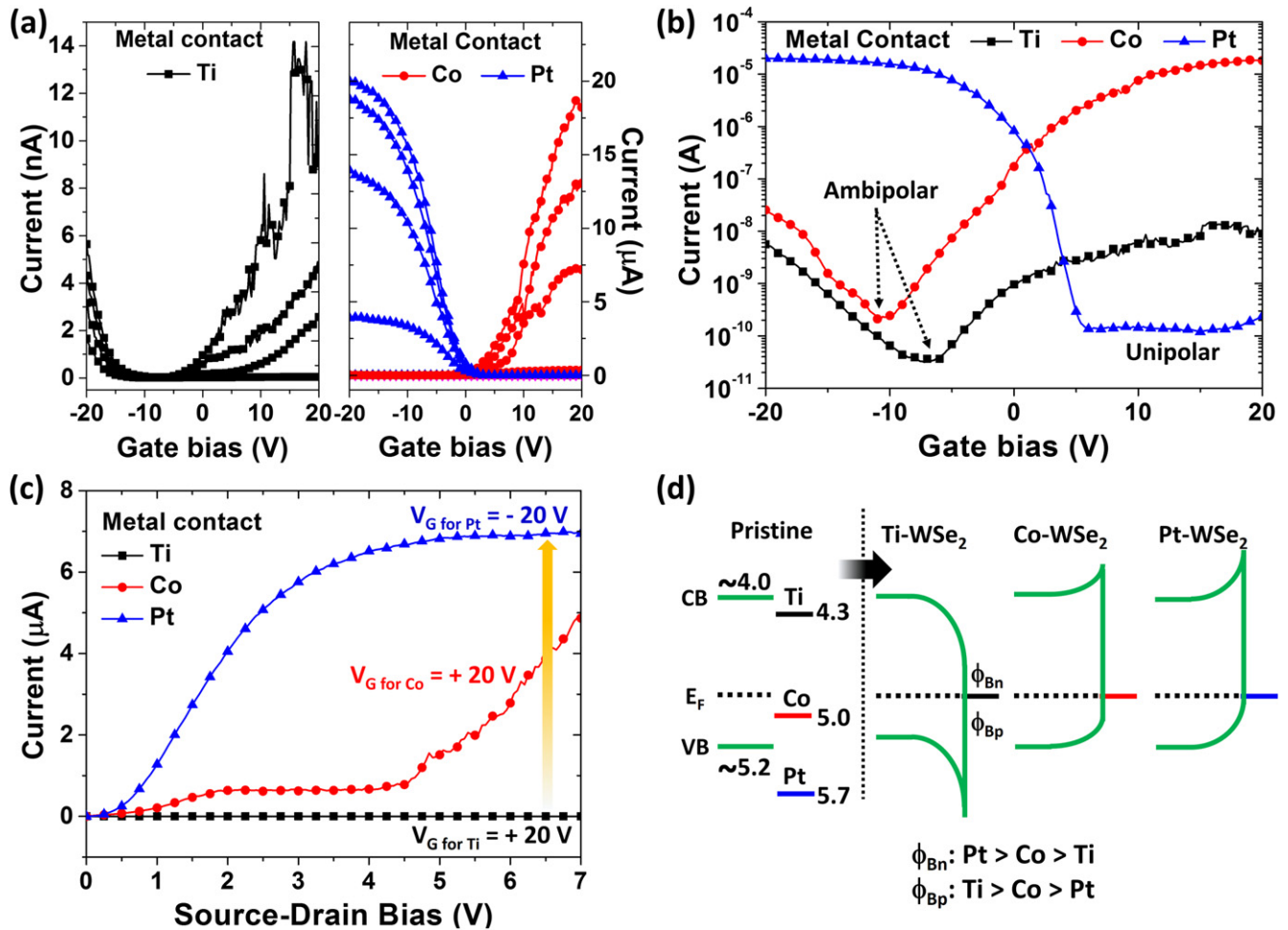
### 3.1. Electrical properties of pristine WSe<sub>2</sub> FETs with different metal contacts

Figures 2(a) and (b) are plots of the transfer characteristics ( $I_D$ – $V_G$ ) of the Ti–, Co–, and Pt–WSe<sub>2</sub> FETs in linear and logarithm scales. It can be seen that the electrical characteristics vary significantly with the contacting metals, which have different work functions of 4.3, 5.0, and 5.7 eV for Ti, Co, and Pt, respectively [13]. The Pt–WSe<sub>2</sub> FET shows a unipolar p-type behaviour with a threshold voltage ( $V_{th}$ ) at 6 V whereas Co–WSe<sub>2</sub> and Ti–WSe<sub>2</sub> FETs show ambipolar behaviour with  $V_{th}$  at –11 and –7 V, respectively. Although the Ti– and Co–WSe<sub>2</sub> FETs show quasi-ambipolar behaviour, the conduction was dominated by electrons rather than by holes. The on-currents of Pt– and Co–WSe<sub>2</sub> were approximately  $10^{-5}$  A, which is three orders of magnitude higher than that of Ti–WSe<sub>2</sub>, as shown in figure 2(c). (For further analysis of  $I_D$ – $V_D$  characteristics, see supplementary information figure S1 and related discussion.) This variation in conduction carrier type and current magnitude can be explained by the difference in the contact (or Schottky) barrier height at the metal–WSe<sub>2</sub> interface [5]. Because TMDC-based 2D materials have unique surface properties, free from dangling bonds and surface states, this clean interface suits the Schottky theory rather than the Bardeen theory, which is based on Fermi level pinning because of surface states and dangling bonds as commonly observed in bulk semiconductors such as silicon- and germanium-based devices [14, 15]. In figure 2(d), band alignments of the conduction and valence bands (CB and VB) of WSe<sub>2</sub> with the work function of metals are indicated [16, 17]. In case of Ti contact, the work function lies near the CB; consequently, a high hole barrier ( $\phi_{bp}$ ) was formed. In contrast, the work functions of Co and Pt metals are closer to the VB, which results in a high electron barrier ( $\phi_{bn}$ ) and relatively low  $\phi_{bp}$ . From the band diagram, Ti–WSe<sub>2</sub> is expected to have low electron/hole conduction whereas the conduction of Co– and Pt–WSe<sub>2</sub> are expected to be dominated by holes. In the case of Ti– and Pt–WSe<sub>2</sub>, the measured electrical results are in agreement with the prediction; however, the Co–WSe<sub>2</sub> shows an unexpectedly high electron conduction. This high electron conduction was also observed in several Co–WSe<sub>2</sub> FETs fabricated separately, suggesting the possibility that the Fermi level of Co metal was pinned near the CB of WSe<sub>2</sub>.

The field-effect mobility was extracted from the linear  $I_D$ – $V_G$  curve at  $V_D = 1$  V, using expression

$$\mu = \frac{L}{W} \left( \frac{dI_D}{dV_G} \right) \left( \frac{1}{C_i \times V_D} \right), \quad (1)$$

where  $W$  and  $L$  are the channel width and length, respectively, and  $C_i = 38$  nF cm<sup>-2</sup> is the capacitance per unit area between the gate and the channel layer. The mobility values are  $3.12 \times 10^{-5}$ , 0.55, and  $9.16$  cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup> for Ti–, Co–, and Pt–WSe<sub>2</sub> FETs, respectively. Because the fabrication conditions and exfoliated layers were similar for all samples, the variation observed in mobility can be attributed



**Figure 2.**  $I_D$ - $V_G$  characteristics for Ti-, Co-, and Pt-WSe<sub>2</sub> FETs in (a) linear scale with  $V_D$  from 1 to 7 V with 2 V steps and (b) logarithm scale with  $V_D = 7$  V. (c)  $I_D$ - $V_D$  characteristics with gate bias at on-current state. (d) Schematic of band diagram for Ti-, Co-, and Pt-WSe<sub>2</sub> FETs before and after band alignment, where  $\phi_{bn}$  and  $\phi_{bp}$  indicate electron and hole contact barriers, respectively. The unit of CB, VB, and  $E_F$  is eV.

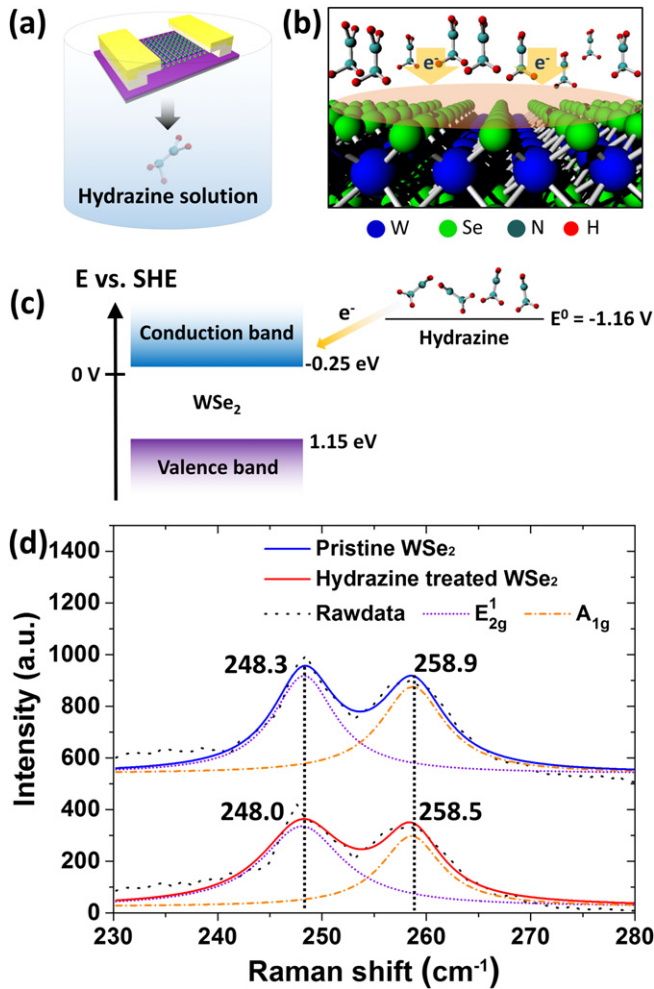
to the effect of the contact resistance. Devices fabricated using Pt contact showed the highest mobility compared to the devices with Ti and Co contacts because Pt exhibited the lowest contact resistance. Consequently, the on/off ratio takes values of  $10^3$ ,  $10^4$ , and  $10^5$  for Ti-, Co-, and Pt-WSe<sub>2</sub> FETs, respectively. From the measured transport properties, Pt and Co metals proved to be a suitable contact for improving the electrical performance of p- and n-type transport in WSe<sub>2</sub> FETs, respectively, whereas the Ti-contacted WSe<sub>2</sub> FET exhibited low electrical performance for both electron and hole transport.

### 3.2. Electrical properties of hydrazine-treated WSe<sub>2</sub> FETs with different metal contacts

Having identified the work function dependence of the contact metal, we investigate the surface doping effect of hydrazine monohydrate on the WSe<sub>2</sub> FET. The Ti-, Co-, and Pt-WSe<sub>2</sub> FETs were dipped in hydrazine solution with a concentration of 80% for 20 s to achieve the n-doped property of the WSe<sub>2</sub> layer, as shown in figure 3(a). Because

hydrazine, also known as diazane, is composed of two amine molecules, it has a strong tendency to donate electrons to the WSe<sub>2</sub> surface, making the channel predominately n-type as shown in figure 3(b). A similar charge transfer mechanism has also been demonstrated in graphene and carbon nanotube-based devices [18, 19]. This doping mechanism can also be verified from the redox potential values. As illustrated in figure 3(c), the CB edge energy of WSe<sub>2</sub> at neutral pH lies at approximately  $-0.25$  eV versus the standard hydrogen electrode, which is much smaller than the standard reduction potential of hydrazine ( $-1.16$  V) [20, 21]. As a result, the electrons from hydrazine are donated to WSe<sub>2</sub>, thereby achieving n-doping in WSe<sub>2</sub> FETs. This doping was further confirmed by the Raman analysis. Figure 3(d) shows the Raman spectra for untreated and treated WSe<sub>2</sub> layers in the hydrazine solution. After hydrazine treatment, the  $E_{2g}^1$  and  $A_{1g}$  modes slightly shifted to a lower frequency with peak positions at  $248.0$  and  $258.5$   $\text{cm}^{-1}$ . This is in good agreement with the previously reported n-doped WSe<sub>2</sub> layer [9], although the shifts are less significant because the material in this study is not mono- or bilayer. In thick TMDC layers, the layers closer





**Figure 3.** Schematics of (a) hydrazine treatment method and (b) the mechanism of the hydrazine-based doping process. (c) Doping mechanism in multi-layered WSe<sub>2</sub>. (d) Raman spectra of the WSe<sub>2</sub> layer before and after hydrazine treatment.

to the surface are more effectively doped compared to the layers far from the surface, which explains the smaller Raman shift observed in our measurement [22].

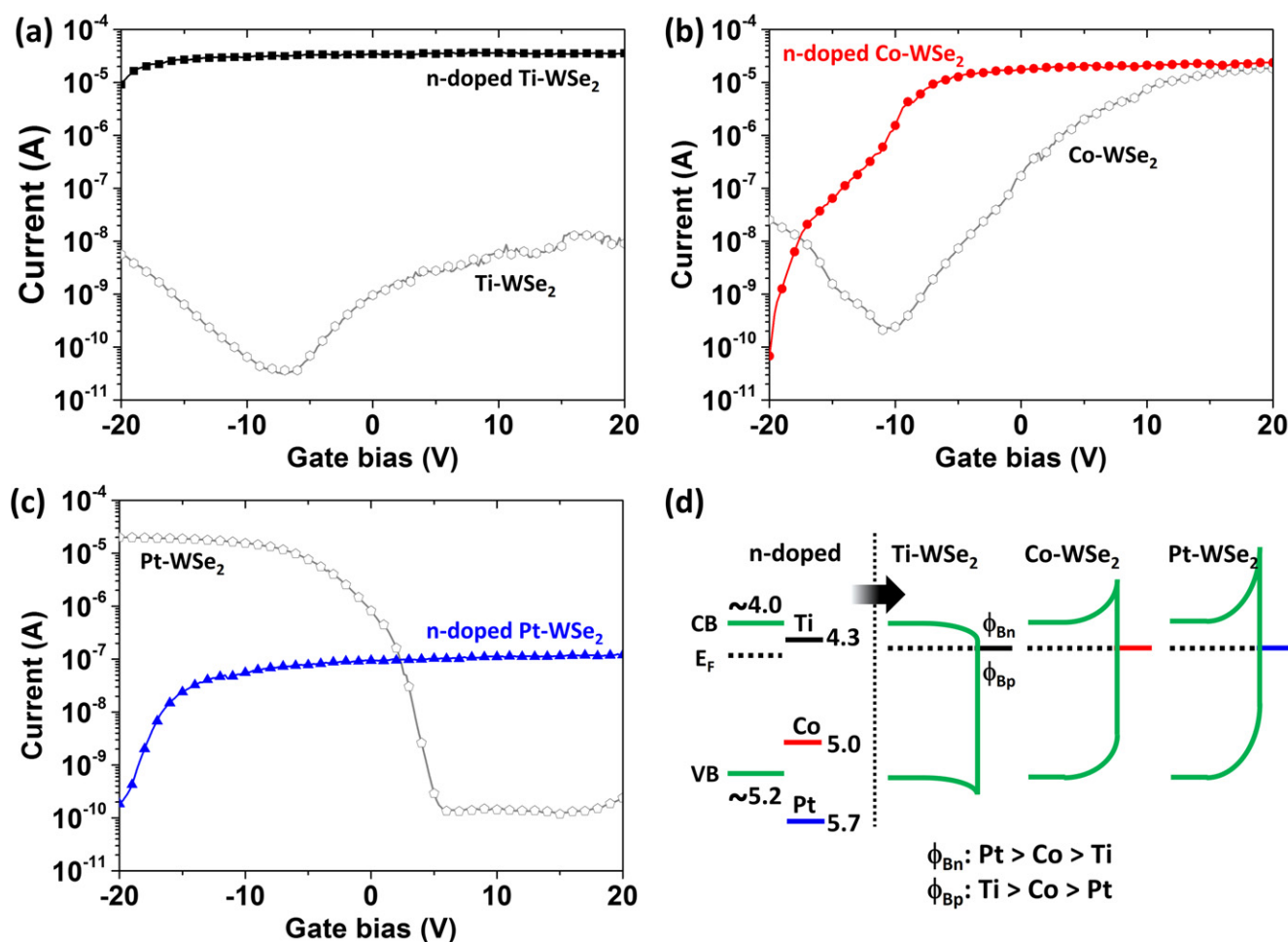
The  $I_D$ - $V_G$  characteristics of hydrazine-treated WSe<sub>2</sub> FETs are shown in figures 4(a)–(c). In the case of the Ti-WSe<sub>2</sub> FET (figure 4(a)), the threshold voltage shifted from -7 to beyond -20 V, indicating a heavily n-doped channel with current and mobility increasing by three orders of magnitude compared to the untreated Ti-WSe<sub>2</sub> FET—i.e., from  $3.12 \times 10^{-5}$  to  $0.01 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ . Similarly, the Co-WSe<sub>2</sub> FET also shows a shift of  $V_{th}$  from -11 to approximately -20 V with roughly same on-current of  $10^{-5} \text{ A}$ , as shown in figure 4(b). However, the mobility of the hydrazine-treated Co-WSe<sub>2</sub> FET slightly decreases from 0.55 to  $0.38 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ . Interestingly, in the case of the Pt-WSe<sub>2</sub> FET (figure 4(c)), the initial p-type WSe<sub>2</sub> FET was converted to n-type after treatment, with a decrease of on-current from  $10^{-5}$  to  $10^{-7} \text{ A}$  and effective mobility from 9.16 to  $0.07 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ . It may be noted that the value of the effective mobility is the combination of the intrinsic mobility and the quality of the ohmic contact; i.e., if the contact

resistance is higher than for the similar intrinsic mobility, the observed field-effect mobility would decrease. Therefore, the increase in the field-effect mobility in the Ti-WSe<sub>2</sub> FET could have its origin from the improvement in the contact resistance after hydrazine treatment.

The observed electrical properties can be explained by the shift of the Fermi level from the vicinity of VB to CB, as shown in figure 4(d). The quasi position of the Fermi level can be calculated by using typical semiconductor carrier density equations [23]. However, in the case of two-terminal measurement, where contact resistance cannot be excluded from the channel resistance, such analysis can lead to erroneous conclusions. In view of this limitation, a qualitative analysis has been employed to study the observed variation in the device characteristics. (For a detail discussion on this issue, see supplementary information, section 2). The up-shift in the Fermi level of WSe<sub>2</sub> due to n-doping treatment lowers the barrier for electron injection, which results in higher on-current and enhanced mobility in the Ti-WSe<sub>2</sub> FET. In the case of Co- and Pt-WSe<sub>2</sub> FETs, a relatively higher  $\phi_{bn}$  was formed, which results in lower on-current and mobility compared to the Ti-WSe<sub>2</sub> FET. In contrast,  $\phi_{bp}$  was highly increased; consequently, the hole transport vanished after the n-doping process. Furthermore, n-type doping in the Ti-WSe<sub>2</sub> FET shows weak degenerate behaviour whereas for Co and Pt contacts, it was possible to modulate the total carrier concentration by gate bias owing to higher barrier height for electron conduction. Finally, the doped devices also showed enhanced hysteresis behaviour, which can be utilized for memory applications. (For details, see supplementary information figure S3 and the related discussion.) From the analyzed electrical measurements, we found that high performance of the n-type WSe<sub>2</sub> FET can be achieved with (1) a low work function (near the CB of WSe<sub>2</sub>) metal contact (Ti in this study), and (2) n-doping by using a hydrazine treatment. The results reveal not only the importance of contact engineering to improve device performance but also the critical role of facile doping in 2D materials, which can extend the device functionality for various applications such as digital and optoelectronics.

#### 4. Conclusion

In conclusion, we studied WSe<sub>2</sub> FETs with various metal contacts using Ti, Co, and Pt and applied a facile hydrazine treatment method to realize high performance of n-type WSe<sub>2</sub> FETs. We analyzed the contribution from the metal work function to the electrical properties before and after n-doping treatment. After the n-doping treatment, the Ti- and Co-WSe<sub>2</sub> FETs showed a highly n-doped behaviour with enhanced electron conduction and negatively shifted  $V_{th}$ , whereas the p-type Pt-WSe<sub>2</sub> was changed to an n-type FET. This doping treatment can be utilized to realize complementary devices and logic circuits based on atomically thin TMDC materials.



**Figure 4.**  $I_D$ - $V_G$  characteristics before and after hydrazine treatment of (a) Ti-, (b) Co-, and (c) Pt-WSe<sub>2</sub> FETs with  $V_D$  at 7 V. (d) Schematic of the band diagram for hydrazine-treated Ti-, Co-, and Pt-WSe<sub>2</sub> FETs before and after band alignment, where  $\phi_{bn}$  and  $\phi_{bp}$  indicate electron and hole contact barriers, respectively. The unit of CB, VB, and  $E_F$  is eV.

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